## Importance of Blocking Layers at Conducting Glass/TiO<sub>2</sub> Interfaces in Dye-sensitized Ionic-liquid Solar Cells

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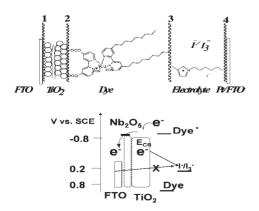
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Thin Nb<sub>2</sub>O<sub>5</sub> film works as a potential blocking layer when deposited between fluorine-doped tin oxide and nanocrystalline  $TiO_2$  layer, improving  $V_{oc}$  and conversion efficiency of the dyesensitized  $TiO_2$  solar cells using ionic liquid electrolytes.

In dye-sensitized  ${\rm TiO_2}$  solar cells (DSCs) that compose of a dye-adsorbed nano- ${\rm TiO_2}$  layer on fluorine-doped tin oxide (FTO) glass cathode as the window electrode, redox electrolytes as charge carrier and the counter anode electrode, unidirectional charge flow with no electron leakage at the interfaces is essential for the high energy-conversion efficiency. Recently, much attention  $^{1-3}$  has been paid to improve the performance of the ionic-liquid DSCs because of their features such as high ionic conductivity, nonvolatility, electrochemical stability, and nonflammability.  $^{4-6}$ 

According to the unidirectional electron-transporting principle of DSCs, there are four important interfaces in the devices as shown in Figure 1. These are, the interface of FTO/TiO2, TiO2/dye, dye/electrolyte, and electrolyte/counter electrode (usually platinized FTO electrode). Recently, many researchers pay much attention to the interface of TiO2/dye especially through core—shell structured electrodes by introducing Nb2O5, SrTiO3, and Al2O3 $^{7-11}$  for TiO2 electrode surface modification. However, except for the establishment of model for DSCs,  $^{12-15}$  few groups investigate the modification of the interface of FTO/TiO2 and its effectiveness by employing a compact TiO2 layer.  $^{12b,15}$ 

In this letter, we first develop  $Nb_2O_5$  as blocking layer which gives an improvement in open-circuit photovoltage  $(V_{\rm oc})$  and fill factor, leading to respectable energy-conversion efficiency in ionic-liquid DSCs. The improvement may come from the  $Nb_2O_5$  forming potential barrier between FTO and



**Figure 1.** Schematic views of interfaces in the DSC device and the electron transfer of the new structured electrode.

TiO<sub>2</sub>, which can suppress back electron transfer from FTO to electrolytes without lowering conductivity between them as shown in Figure 1.

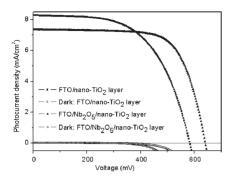
The spray pyrolysis method  $^{12a}$  with different precursors listed in Table 1 has been applied to form the corresponding metal oxide layers on FTO during the fabrication of the structured FTO/metal oxide/porous TiO<sub>2</sub> (Nanoxide-T, Solaronix, 5.5  $\mu$ m) electrodes. The general procedure is as follows; each ethanol solution (0.02 M) of the precursor was sprayed layer-by-layer appropriate repetitions onto the FTO substrate keeping over  $400\,^{\circ}$ C. Then, the substrates with the blocking layers were annealed at 500  $^{\circ}$ C for 1 h (some XRD data refer to supplementary information). The Ru dye, Z-907 (Ru-520 DN, Solaronix) and the ionic-liquid electrolyte composed of HMImI (1-hexyl-3-methylimidazolium iodide) with iodine ([I<sup>-</sup>]:[I<sub>2</sub>] = 10:1) are employed for ionic-liquid DSCs. The cells without any blocking layer or with liquid electrolyte based on methoxyacetonitrile are also fabricated for comparison.

As shown in Table 1, while the blocking MgO, ZnO,  $Al_2O_3$ ,  $Eu_2O_3$ , and  $SiO_2$  layers do not give any improvement of  $V_{oc}$ , the Nb<sub>2</sub>O<sub>5</sub> blocking layer improves  $V_{oc}$  effectively, with keeping respectable fill factors (FF), giving the high conversion efficiency as well as the TiO<sub>2</sub> blocking layer. Figure 2 presents the J-V curve of the novel type solar cell and that of nonblocked reference under AM 1.5 irradiation of  $100 \, \mathrm{mW/cm^2}$ .

**Table 1.** Comparison of the parameters of the ionic-liquid DSCs using mesoporous TiO<sub>2</sub> electrodes with several metal oxide blocking layers on FTO after optimization

Precursors	Blocking	$V_{ m oc}$	$J_{ m sc}$	FF	$\eta/\%$
	materials	/mV	/mA		1// /0
_	none	580	8.30	0.57	2.8
a	$TiO_2$	641	7.85	0.65	3.3
Nb(OCH <sub>2</sub> CH <sub>3</sub> ) <sub>5</sub>	$Nb_2O_5$	649	7.43	0.67	3.2
$Mg(OCH_3)_2$	MgO	590	7.38	0.64	2.8
Sn(CH <sub>3</sub> COCHCO	$SnO_2$	578	8.63	0.57	2.8
$CH_3)_4$	$SHO_2$	376	0.03	0.57	2.6
Zn(CH <sub>3</sub> COCHCO	ZnO	572	4.44	0.54	1.4
$CH_3)_2$	Ziio	312	7.77	0.54	1.7
$Al(OCH(CH_3)_2)_3$	$Al_2O_3$	579	5.35	0.69	1.8
EuCl <sub>3</sub>	$Eu_2O_3$	621	6.06	0.61	2.3
Tetramethylcyclo-	SiO <sub>2</sub>	567	6.75	0.57	2.2
tetrasiloxane	3102	307	0.73	0.57	2.2
_	none <sup>b</sup>	704	10.17	0.68	4.9
$Nb(OCH_2CH_3)_5$	$Nb_2O_5^b$	710	9.54	0.72	4.8

<sup>&</sup>lt;sup>a</sup>Diisopropoxytitanium bisacetylacetonate as precursor. <sup>b</sup>Liquid electrolyte: 0.1 M LiI, 0.3 M 2,3-dimethyl-1-propylimidazolium iodide, 0.05 M  $\rm I_2$ , 0.5 M 4-*tert*-butylpyridine in methoxyacetonitrile.



**Figure 2.** J-V curves of cells employing Z-907 sensitized FTO/nano-TiO<sub>2</sub> ( $\blacksquare$ ) and FTO/Nb<sub>2</sub>O<sub>5</sub>/nano-TiO<sub>2</sub> electrodes ( $\blacktriangle$ ) under AM 1.5 irradiation. (Electrolyte:HMImI:I<sub>2</sub> = 10:1).

When the thicker Nb<sub>2</sub>O<sub>5</sub> layer (obtained by 50-times pyrolyses, being estimated as around  $\approx$ 50 nm) is employed, the device gives a great improvement of  $V_{\rm oc}$  about 70 mV and better fill factor with the compensation of some lower  $J_{\rm sc}$  and finally gives a 3.3% energy-conversion efficiency which is 15% higher than that of the reference. The great improvement of  $V_{\rm oc}$  can be explained by the diode equation: <sup>16,17</sup>

$$V_{\rm oc} = (nRT/F) \ln[(J_{\rm sc}/J_0) - 1]$$
 (1)

where n is the ideality factor whose value is between 1 and 2,  $J_{sc}$ and  $J_0$  are the photocurrent density under short circuit and reverse saturation current, respectively, R and F have the common meaning. The literature 16 suggested that charge recombination at the nanocrystalline/redox electrolyte interface is expected to play a significant role in lowering the photovoltage. Acceptably, there are two recombination pathways occurring at the interface. The injected conduction-band electrons recombine with oxidized dye or triiodide and polyiodide redox species in the electrolyte. And usually the former reaction is ignored because of the rapid rate of reduction of oxide dye molecules by I<sup>-</sup> ions. Many attempts have been done to suppress such back transfer reaction taken place at the surface of TiO<sub>2</sub> in the last several years using surface treatment of TiO<sub>2</sub> electrode.<sup>7-11</sup> On the other hand, the porous interfaces between FTO substrate and TiO2 can also act as electron recombination sites, i.e., electron leakage sites especially in solid- or quasi-solid-state electrolytes like ionicliquid electrolytes. Our newly structured FTO/Nb<sub>2</sub>O<sub>5</sub>/TiO<sub>2</sub> electrode that should form 100-mV18 potential barrier can prohibit the recombination of injected electron in FTO with the redox electrolyte effectively, which means the enhancement of the electron collection at FTO, giving an improvement of shunt resistance and  $V_{oc}$  for the accumulation of injected

On the other hand, the improved fill factor maybe due to the lower series resistance.  $^{19}$  Such an explanation is supported by the evidence that the sheet resistance of Nb<sub>2</sub>O<sub>5</sub>-doped SnO<sub>2</sub> is decrease compared with SnO<sub>2</sub> after high temperature calcination.  $^{20}$ 

It should be noted that such kinds of structured electrode have less influence in liquid-electrolyte system and that this assures the fluid solvent molecules can easily penetrate into the porous FTO/TiO<sub>2</sub> interfaces, working as separators from the redox species. Since in solid state DSCs, or ionic-liquid DSCs, it is not easy to change the surface potential of TiO<sub>2</sub> by

adding additives like TBP (4-tert-butylpyridine) to improve the  $V_{\rm oc}$  of the cell, such kind of modified substrate is a good choice for the improvements of  $V_{\rm oc}$ . Other facile methods for construction of the effective blocking layer are currently being undertaken

In summary, we have demonstrated that among the examined metal oxides,  $Nb_2O_5$  can form an effective blocking layer at FTO/nano-TiO<sub>2</sub> interfaces and greatly improve  $V_{oc}$  and fill factor, proving the importance of blocking layer at FTO and TiO<sub>2</sub> interfaces in ionic-liquid DSCs.

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